Refining Energy Levels in ReS₂ Nanosheets by Low-Valent Transition-Metal Doping for Dual-Boosted Electrochemical Ammonia/Hydrogen Production

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Electrocatalytic nitrogen reduction reaction (NRR) and hydrogen evolution reaction (HER) are intriguing approaches to nitrogen fixation and hydrogen production under ambient conditions, given the need to discover efficient and stable catalysts to light up the "green chemistry" future. However, bottlenecks are often found during N_2/H_2O activation, the very first step of NRR/HER, due to energetic electron injection from the surface of electrocatalysts. It is reported that the bottlenecks for both NRR and HER can be tackled by engineering the energy level via low-valent transitionmetal doping, simultaneously, where rhenium disulfide (ReS₂) is employed as a model platform to prove the concept. The doped low-valent transitionmetal domains (e.g., Fe, Co, Ni, Cu, Zn) in ReS₂ provide more active sites for N₂/H₂O chemisorption and electron transfer, not only weakening the $N \equiv N/O-H$ bonds for easier dissociation through proton coupling, but also elevating *d*-band center toward the Fermi level with more electron energy for N₂/H₂O reduction. As a result, it is found that iron-doped ReS₂ nanosheets wrapped nitrogen-doped carbon nanofiber (Fe-ReS2@N-CNF) catalyst exhibits superior electrochemical activity with eightfold higher ammonia production yield of 80.4 μ g h⁻¹ mg⁻¹_{cat}, and lower onset overpotential of 146 mV and Tafel slope of 63 mV dec⁻¹, when comparing with the pristine ReS₂.

(large bond energy of 940.95 kJ mol⁻¹),^[6,7] makes it a "Cadmean victory" to convert N₂ into NH₃. For instance, the famous reaction for industrial ammonia production is known as Haber–Bosch process, relying on iron-based catalysts and rigorous operating conditions (673–873 K, 20–40 MPa),^[8,9] which almost accounts for 1% of the worldwide energy consumption and 1.6% of the total global CO₂ emission.^[10]

In order to promote the progress of "green chemistry", a clean, safe, and sustainable NH₃ synthesis route is an urgent need. Electrochemical nitrogen reduction reaction (NRR) has gradually attracted much attention, and been regarded as an alternative approach.^[11–14] As an inevitably overwhelming rival, the hydrogen evolution reaction (HER) should be well impeded, making it a key issue to find a reasonable way to enhance N species adsorption on the surface of catalyst instead of H species. For instance, Ling et al. put forward a reticular chemistry

1. Introduction

Ammonia (NH₃) is an indispensable carbon-free chemical to agriculture and other products,^[1–3] as well as a potential carrier for hydrogen storage due to its high hydrogen density and lique-fiable feature.^[4,5] However, the chemical inertness of dinitrogen

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approach by exploiting MOF's water-repelling and molecularconcentrated effects to overcome HER-imposed bottlenecks (MOF: metal–organic framework), achieving both excellent NRR selectivity (≈90%) and boosted Faradaic efficiency by 10 percentage points.^[15] Recently, Yan et al. reported a strategy to simultaneously promote NRR selectivity and activity by using

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bismuth nanocrystals and potassium cations, the Faradaic efficiency of which was achieved as 66%.^[16] Yan et al. also proposed it can suppress the HER process and improve the Faradaic efficiency of NRR to 56.55% by positively shifting the reaction potential and using single-atom iron as electrocatalyst.^[17] To our knowledge, however, traditional Haber-Bosch process will still be the main route for NH₃ production in the foreseeable future with a large amount usage of H₂ by steam reforming of fossil fuels, where HER would be a more attractive choice for renewable H₂ resources. In fact, the majority used hydrogen gas for Haber-Bosch process would go through a series of chemical treatment processes under high temperature, while less than 4% of the total H₂ gas comes from sustainable and mild electrocatalysis of water. Therefore, it would be also another possible perspective to explore dual-enhanced electrocatalysts for both ammonia and hydrogen production, as it is still difficult to suppress the HER process well for the majority electrocatalysts.

Bearing this in mind, rhenium disulfide (ReS2) has been recently discovered as next-generation transition metal dichalcogenide (TMD) materials for high-efficient electrocatalysis.^[18,19] It is mainly attributed to their layered structure with weak van der Waals interactions between individual sandwiched layers,^[20,21] and anisotropic structure with two-dimensional (2D) morphology to provide not only large specific surface area but also 2D permeable channels for reactant adsorption/transport.^[22] In the past few years, abundant effort has been made to promote the development of TMD-based electrocatalysts for both nitrogen reduction reaction and hydrogen evolution reaction.^[23-25] In contrast with other TMD materials (such as MoS₂, MoSe₂, WS₂, and WSe₂), ReS₂ possesses a very weak interlayer coupling,^[25–27] thus leading to an enhanced reactant diffusion into its deep interlayer space for better NRR/HER processes. Besides that, traces of transition-metal doping, particularly for low-valent domains (such as Fe³⁺, Co²⁺, Ni²⁺, Cu²⁺, Zn²⁺), $[2^{(2-30)}]$ in atomically ordered ReS₂ nanosheets can disturb its crystalline lattice to expose more active sites for electrocatalysis. However, several significant issues in low-valent transition-metal doped ReS₂ nanosheets are still rarely reported and urgently needed to be addressed and solved, namely, how does transition-metal incorporate into ReS₂, and where are the real electrocatalytic sites for NRR/HER?

Therefore, here we employ low-valent Fe³⁺/Co²⁺/Ni²⁺/Cu²⁺/ Zn^{2+} salts deliberately for coprecipitation with ReO_4^- salt. By this, M4+-positioned Re could be replaced by low-valent transition-metal species (e.g., Fe, Co, Ni, Cu, Zn). The density functional theory (DFT) calculation first reveals that the energy level of ReS₂ can be well tuned by doping engineering with enhanced charge density around the low-valent cations. Taking iron-doped ReS₂ nanosheets wrapped nitrogen-doped carbon nanofiber (Fe-ReS₂@N-CNF) catalyst as an example, it exhibits superior electrochemical activity with an eightfold higher ammonia production yield rate of 80.4 μ g h⁻¹ mg⁻¹_{cat.} for NRR, and lower onset overpotential of 146 mV and Tafel slope of 63 mV dec⁻¹ for HER, when comparing with the pristine ReS₂. The deep mechanisms for NRR/HER processes are further discovered by in situ spectroscopic characterization together with DFT studies that low-valent Fe domain is beneficial to providing more active sites for N₂/H₂O chemisorption and weakening the N≡N/O-H bonds for easier dissociation through proton coupling. These results clearly demonstrate the important role of low-valent transition-metal doping engineering on optimizing ReS2-based NRR/HER electrocatalyst, which provides a novel but general route for green synthesis of NH₃ production.

2. Results and Discussion

Foreseen by computational study first, the pristine ReS_2 semiconductor shows a wide energy gap (E_g) of 1.31 eV (**Figure 1**a)

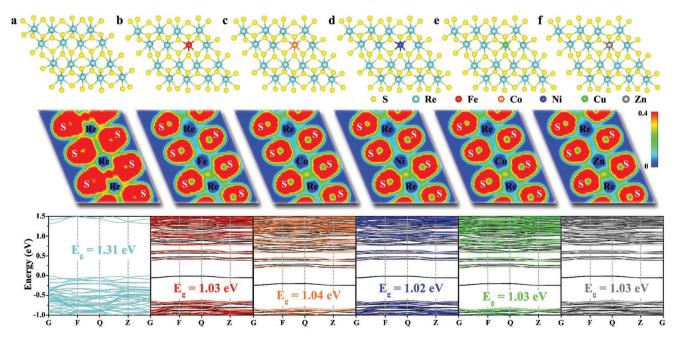


Figure 1. Atomistic models of crystal structure, two dimensional electron localization function (ELF) mappings, and calculated band structures for a) ReS₂, b) Fe-ReS₂, c) Co-ReS₂, d) Ni-ReS₂, e) Cu-ReS₂, and f) Zn-ReS₂, respectively.



as direct-band gap. This high $E_{\rm g}$ will certainly hinder the catalyst activation with inefficient electron transition. Interestingly, the other five kinds of optimized ReS₂ models were constructed by doping 5% low-valent transition-metals (e.g., Fe, Co, Ni, Co and Zn), denoted as Fe-ReS₂ (Figure 1b), Co-ReS₂ (Figure 1c), Ni-ReS₂ (Figure 1d), Cu-ReS₂ (Figure 1e), and Zn-ReS₂ (Figure 1f), respectively. As a result, we found the E_{g} values of these doped ReS₂ samples significantly decreased to around 1.03 eV as indirect-band gaps, demonstrating it a feasible strategy for superior electron excitation of transition-metal doped ReS₂. The well-reduced energy gap is attributed to the impurity band after doping of transition-metals, which can be either in the gap or in the valence/conduction band. Here, the first-principle calculations show that impurity band will live in the gap of transition-metal doped ReS₂, therefore the gap is decreased. Meanwhile, this may also be originated from the activate ReS₂ atomic surface with charge redistribution, especially centered around low-valance transition-metal doped domains (Figure 1 and Figure S1, Supporting Information),^[31] which can be regarded as highly active electrocatalysts.

Inspired by the above theoretical results, iron-doped ReS₂ nanosheets wrapped nitrogen-doped carbon nanofiber (Fe-ReS2@N-CNF) is first chosen as the key model system to investigate the benefits of low-valent transition-metal doping engineering. The samples are synthesized via an N-CNF templated hydrothermal method,^[32] in which Fe atoms are doped into ReS₂ lattice using FeCl₃·6H₂O as a precursor as schematically illustrated in Figure 2a. First, the typical scanning electron microscopy (SEM) image of N-CNF is displayed in Figure 2b with average diameter of 30–60 nm, proving that the heteroatom doping process for nitrogen atoms would not affect its morphology (Figure S5, Supporting Information) and noncrystalline property (Figure S6, Supporting Information). The nitrogen-rich carbon surface is also beneficial to the adsorption of ReO₄⁻/Fe³⁺ precursors for epitaxial growth of Fe-ReS₂ nanosheets. Figure 2c-e reveals the morphology of Fe-ReS2@N-CNF by SEM and transmission electron microscopy (TEM), highlighting a feature of uniform Fe-ReS₂ nanosheets covered hybrid carbon nanofiber with ultrafine diameter of ≈85 nm and length extending to several micrometers. This hybrid strategy is beneficial to solving the difficult problem of Fe-ReS₂ aggregations (Figure S7, Supporting Information), leading to significantly increased specific surface area from 12.1 to 24.6 cm² g⁻¹ (Figure S8, Supporting Information). As displayed in highresolution TEM (HRTEM) image (Figure 2f), the iron-doped ReS₂ nanosheets with a spacing of 0.62 nm can be assigned to (001) planes, indicating that the Fe-ReS₂ mainly grow along [001] direction. Energy-dispersive X-ray spectroscopy (EDS) mapping is further employed to ascertain the doping of Fe and N elements on Fe-ReS₂@N-CNF-5, revealing the uniform Fe and N doping (Figure 2g). X-ray diffraction (XRD) exhibits the same Bragg reflections of Fe-ReS2@N-CNF-5 and non-doped ReS₂@N-CNF (Figure 2h), implying no structural deformation caused by low-valent Fe substitution of Re⁴⁺.

X-ray photoelectron spectroscopy (XPS) is further applied to understand the chemical states and elemental compositions in the Fe-ReS₂@N-CNF-5. The binding energies of the Re⁴⁺ $4f_{7/2}$ and $4f_{5/2}$ electron peaks in Fe-ReS₂@N-CNF-5 are 41.0 and 44.4 eV, respectively (**Figure 3**a).^[33] As shown in Figure 3b,

two characteristic peaks being located at 162.4 and 163.4 eV in Fe-ReS₂@N-CNF-5 are detected for the bonding configurations of S, corresponding to the core $2p_{3/2}$ and $2p_{1/2}$ level peaks of S^{2–}, respectively.^[34] All these above results are consistent with values for ReS2@N-CNF, indicating the immeasurable changes for chemical states of Re and S elements after doping iron atoms in Fe-ReS₂@N-CNF-5. Three types of pyridinic N (398.4 eV), pyrrolic N (400.3 eV), and graphitic N (401.1 eV) are identified from Fe-ReS2@N-CNF-5 (Figure 3c), manifesting the successful incorporation of N atoms in the carbon nanofiber template. Meanwhile, the high pyridinic N content in Fe-ReS₂@N-CNF-5 can also promote the following ammonia production.^[35] The binding energies of the Fe $2p_{3/2}$ and Fe $2p_{1/2}$ electron peaks in Fe-ReS₂@N-CNF-5 are 708.1 and 721.2 eV (Figure 3d), respectively, confirming the successful incorporation of Fe atoms in ReS₂ structure. In this instance, an obviously negative shift of binding energy for the Fe $2p_{3/2}$ electron is observed when compared with other iron compounds, such as binding energies of Fe 2p_{3/2} in Fe₂O₃ (710.1 eV), FeCl₂ (710.7 eV), and FeCl₃ (711.6 eV).^[36] It can be ascribed to the bonding hybridization of Fe atom with its neighbor atoms, leading to the decreased oxidation state of Fe. After incorporating low-valent Fe element in ReS₂ nanosheets, the density distribution in Fe-ReS₂ is significantly destabilized to Fe-doped domains (inset in Figure 3d), due to the structural heterogeneity-leaded charge rearrangement. Furthermore, the quantitative analysis of XPS spectra (Figure S9, Supporting Information) demonstrates the chemical formula of 5 mol% Fe-doped ReS2 nanosheets can be fixed to Fe_{0.073}Re_{0.991}S₂ (Table S1, Supporting Information).

Upon confirming the electronic structure of low-valent Fe doped ReS₂ compound, we are now in a position to evaluate the efficacy of as-obtained 5 mol% transition-metal doped ReS₂ nanosheets wrapped nitrogen-doped carbon nanofiber (Fe/Co/ Ni/Cu/Zn-ReS₂@N-CNF-5) for electrochemical HER/NRR performance, which were tested under 0.1 M Na₂SO₄ solution with continuous N2 bubbling. As shown in the polarization curves (Figure 4a) and Tafel plots (Figure 4b), the low-valent transition-metal doping could moderately improve the HER activity of non-doped ReS2@N-CNF (Table S2, Supporting Information). Taking Fe-ReS₂@N-CNF as an example, it shows lower onset overpotential ($\eta = 146$ mV, the onset overpotential is defined as the difference between the reversible potential of the couple H_2/H^+ and the potential where electrocatalysis occurs) and Tafel slope (63 mV dec⁻¹) than those in ReS₂@N-CNF (η = 200 mV; 87 mV dec⁻¹), implying its superiority to accelerate the hydrogen evolution reaction by low-valent transition-metal doping engineering. As an efficient conductive-frame, nitrogen-doped carbon nanofiber is also beneficial for the electron transfer along the interface of N-CNF and ReS₂ outer layer, leading to the enhanced HER performance (Figure S12, Supporting Information). Fe-ReS₂@N-CNF-5 also shows remarkable durability (Figure S13, Supporting Information), without dramatic change of the LSV curve after 5000 cycles. Apart from positive effect of low-valent transition-metal doping engineering in HER activity, we also found it is useful in the improvement of NRR. As shown in Figure 4c, the maximum NH₃ yield rate (80.4 µg h⁻¹ mg⁻¹_{cat.}) and Faradaic efficiency (FE = 12.3%) of Fe-ReS2@N-CNF-5 are much higher than those of non-doped ReS₂@N-CNF (10.3 μ g h⁻¹ mg⁻¹_{cat},





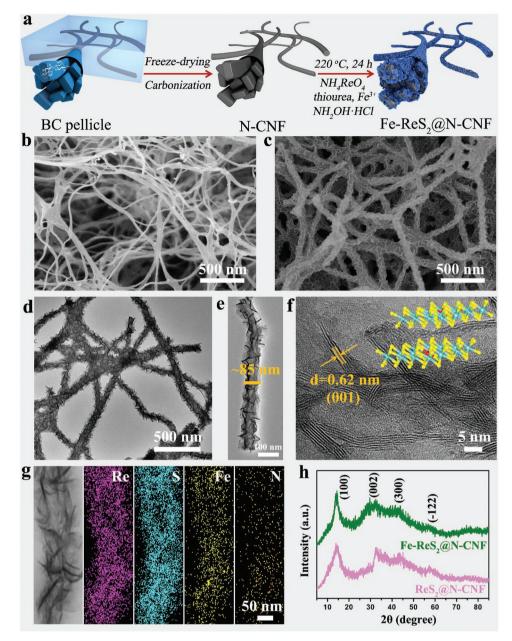
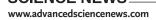


Figure 2. a) Schematic illustration of preparation process of Fe-ReS₂@N-CNF. SEM images of b) N-CNF and c) 5 mol % Fe-doped ReS₂@N-CNF (Fe-ReS₂@N-CNF-5). d,e) TEM images, f) HRTEM image, and g) corresponding element mapping of Fe-ReS₂@N-CNF-5. h) XRD patterns of Fe-ReS₂@N-CNF-5 and ReS₂@N-CNF.

FE = 9.9%) at -0.2 V versus reversible hydrogen electrode (RHE). In order to make the quantitative method for NH₃ yield rate more convincing, nuclear magnetic resonance (NMR) method was further applied (Figure S20a, Supporting Information). Based on the corresponding fitted curve (Figure S20b, Supporting Information), the NH₃ yield rate can be calculated to be 79.8 μ g h⁻¹ mg⁻¹_{cat}, which approached to the NH₃ yield rate from indophenol blue method within the allowed error range, demonstrating the reliability of detection methods (NMR and indophenol blue methods). This enhanced NRR performance can also be observed in other low-valent transition-metal doped electrodes of Co/Ni/Cu/Zn-ReS₂@N-CNF-5 (Figure S21

and Table S3, Supporting Information). To verify the source of ammonia production, NMR method was conducted as ¹⁵N isotopic labeling experiments. As displayed in Figure S22, Supporting Information, only ¹⁴N triplets and ¹⁵N doublets were observed in the ¹⁴NH₄⁺ and ¹⁵NH₄⁺ productions, respectively, by using ¹⁴N₂ and ¹⁵N₂ as feeding gases, which demonstrate not only the purity of as-used ¹⁴N₂ gas but also the stable structure of Fe-ReS₂@N-CNF-5 electrocatalyst without escaped N atoms during NRR process. Meanwhile, it was revealed that the Fe-doping concentrations would be closely related with the corresponding NRR performance (especially for NH₃ yield), 5% Fe-doping amount would be a more proper concentration



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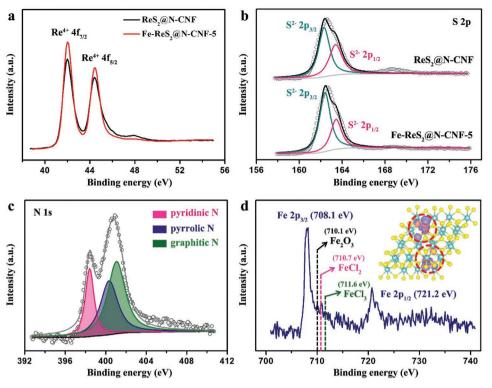


Figure 3. High-resolution XPS spectra of a) Re 4f and b) S 2p peaks for Fe-ReS₂@N-CNF-5 and ReS₂@N-CNF, respectively. High-resolution XPS spectra of c) N 1s and d) Fe 2p for Fe-ReS₂@N-CNF-5.

to release the best electrocatalytic activity for Fe-ReS₂ samples (Figure S24, Supporting Information). Both the NH₃ yield rates and FE_{NH3} values for Fe-ReS₂@N-CNF-5 are also well maintained with a stability retaining of ≈99% after ten continuous electrochemical NRR tests (Figure 4d). From the corresponding XPS spectra (Figure S25, Supporting Information), the Fe-ReS2@N-CNF-5 still maintains the same chemical states and elemental compositions after NRR stability test as the ones before electrocatalysis (Figure 3), demonstrating its excellent electrocatalytic stability for long-term usage. As shown in Figure S26, Supporting Information, the sum of the amounts for generated NH₃ production (S2) and H₂ production (S1) equals the total supplied charge amount (S1 + S3), which indicates all the supplied charge was used to reduce neither N2 gas nor H2O molecules with a Faradaic efficiency of 87.7% for HER process. The rough agreement of the charge amount (S2 \approx S3) between experimentally quantified gases (NH₃ and H₂) and calculated one suggests the total FE is 100% for the sum of NRR and HER without hydrazine generation (Figure S28, Supporting Information).

These enhanced NRR and HER performances in Fe-ReS₂@N-CNF-5 can be attributed to the refined energy levels of ReS₂ nanosheets after low-valent transition-metal doping, as predicted by DFT calculations in Figure 1. Experimentally, ultraviolet photoelectron spectroscopy (UPS) was used to determine the values of energy level (difference between the vacuum and Fermi level) of Fe-ReS₂ and pure ReS₂ as 6.11 and 7.27 eV (Figure 4e and Figure S29, Supporting Information), respectively, by subtracting the width of the He I UPS spectra at excitation energy of 21.22 eV. Compared with the energy level of non-doped ReS₂ nanosheets, the one

of Fe-ReS₂ is properly shifted to approach the reduction potentials of H₂O and N₂ (Figure 4f), making the electron exchange process (Figure S30, Supporting Information) easier between electrocatalyst and the reacting species.^[30] Due to the increased energy levels of Fe-ReS₂ nanosheets, the energy required for electron removal from the catalyst surface decreases, leading to the thermodynamically activated NRR and HER. In another perspective, the feasibility of electron transfer in low-valent transition-metal doped ReS2 electrocatalysts would deliver lower onset overpotentials, demonstrating Fe-ReS2@N-CNF-5 as a promising electrocatalyst for NRR and HER processes. Additionally, the incorporation of highly conductive N-CNF template is beneficial to increasing the electron transfer between two components in the Fe-ReS2@N-CNF-5 for efficient reduction reaction as compared with that in Fe-ReS₂ bulk (Figure S31, Supporting Information).

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Apart from the perspective of thermodynamics, kinetics also plays a key role in these reactions. Attributing to N-CNF template and low-valance Fe doping, Fe-ReS₂@N-CNF-5 displays an increased number of electroactive sites (evidenced by the N₂ adsorption-desorption isotherms in Figure S8, Supporting Information, and optimized electronic structure in Figure 3d), leading to the optimal activation effect of H₃O⁺ and N₂ reagents on catalyst surface. For the HER process (Figure 5a), the H₃O⁺ molecule interacts between Re^{δ +}/Fe^{δ +}···O and S^{δ -}···H synergistically, as the first volmer step (H₂O + e⁻ \rightarrow H_{ads} + OH⁻). After the adsorption of another H₂O molecule, the H₂ gas generates by following one kind of desorption steps (H_{ads} + H₂O + e⁻ \rightarrow H₂ + OH⁻ or H_{ads} + H_{ads} \rightarrow H₂). As revealed by DFT calculation further (Figure S32, Supporting Information), the free

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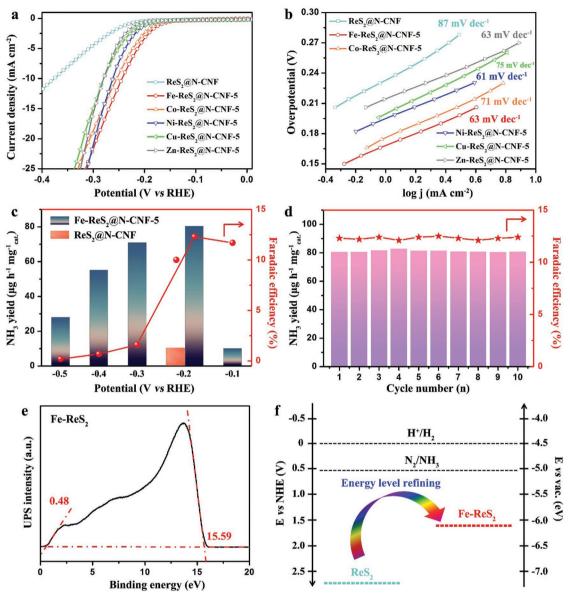


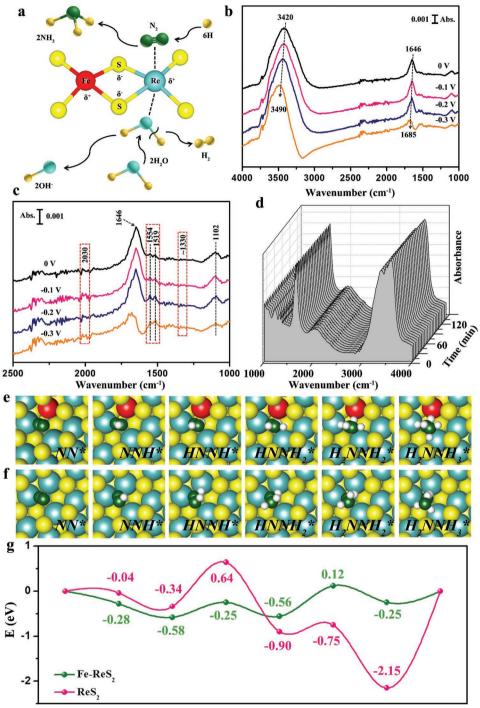
Figure 4. Electrocatalytic HER/NRR of 5 mol% transition-metal-doped ReS_2 nanosheets wrapped nitrogen-doped carbon nanofiber (Fe/Co/Ni/Cu/ Zn-ReS₂@N-CNF-5) and non-doped ReS_2 @N-CNF catalysts in aqueous solution of 0.1 M Na₂SO₄ under ambient conditions. a) Polarization curves. b) Tafel curves. c) The NH₃ production rates (bar graph) and FE_{NH3} (red line) of Fe-ReS₂@N-CNF-5 and ReS₂@N-CNF at different potentials. d) NRR stability test of Fe-ReS₂@N-CNF-5 at -0.2 V versus RHE. e) UPS spectra of Fe-ReS₂ nanosheets. f) Band structure diagram for Fe-ReS₂ and pure ReS₂.

energy value for H* (where * denotes as an adsorption site) is -0.36 eV for the pure ReS₂, but the water adsorption energy is up to 0.81 eV, which lead to the sluggish HER kinetics. Although the free energy value for H* in Fe-ReS₂ is relatively higher to be -0.49 eV, its water adsorption energy can be drastically reduced to 0.29 eV, indicating it is thermodynamically favorable for water reduction on Fe-ReS₂ in neutral solution. To further investigate the nitrogen reduction reaction process (Figure 5a), in situ attenuated total reflectance infrared (ATR IR) spectroscopy measurements were carried out for Fe-ReS₂@N-CNF-5 sample (Figure 5b-d). The IR bands at about 3420 and 1646 cm⁻¹ (Figure 5b) match well with the O-H stretching and H-O-H bending of water molecules, respectively. Their

positive shift with the decrease of applied potential from 0 to -0.3 V is attributed to the change of the adsorption configuration of water.^[37] The peaks at 1554/1519, 1330 and 1102 cm⁻² are attributed to the H–N–H bending,^[38,39] –NH₂ wagging,^[13,40] and N–N stretching^[41,42] of adsorbed N₂H_y species, respectively. These results demonstrate that the nitrogen reduction reaction on Fe-ReS₂ follows an associative mechanism, instead of dissociative mechanism in the Haber-Bosch process,^[43] with the simultaneous addition of hydrogen on N₂ to break the N≡N bond. The time-dependent SEIRAS spectra (Figure 5d) of Fe-ReS₂@N-CNF-5 catalyst at -0.2 V show similar absorption results with time goes, demonstrating its ultra-stable electrocatalysis. As reported in previous literature, enzymatic mechanism







Reaction coordinate

Figure 5. a) The proposed mechanism on uncoordinated Fe (δ^+) and Re (δ^+) centers during NRR and HER processes. b) In situ ATR IR spectra from 0 to -0.3 V on the Fe-ReS₂@N-CNF-5 catalyst in an N₂-saturated 0.1 M Na₂SO₄ solution. c) Amplification of the selected area in (b). d) Time-dependent ATR IR spectra of Fe-ReS₂@N-CNF-5 catalyst at a voltage of -0.2 V. Optimized structures of NH*, NNH*, HNNH*, HNNH₂*, H₂NNH₂*, and H₂NNH₃ intermediates on: e) Fe-ReS₂ and f) ReS₂. g) Free energy diagram of NRR on Fe-ReS₂ and ReS₂.

of N_2 fixation is more suitable for single-atom-embedded systems with lower energy barrier.^[44,45] Therefore, the possible NRR mechanism is depicted in Figure S33, Supporting Information, with DFT calculation results in Figure 5e–g. All

the reaction intermediates were bounded simultaneously to the surface Re atoms of both Fe-ReS₂ and ReS₂, while the Re atoms neighboring Fe atom showed higher activity for the reactions of N-related species (Figures 5e,f). The calculation results further



indicate the active sites in Fe-ReS₂ are located in the region near doped Fe atoms (especially for the Re atoms), which is corresponded with the ELF mappings (Figure 1) and destabilized density distribution (inset in Figure 3d). Further revealed by the free energy diagram (Figure 4g), the adsorption energy of nitrogen on Fe-ReS₂ (-0.28 eV) is much smaller than that on ReS_2 (-0.04 eV), which demonstrates that nitrogen is preferred to be adsorbed and activated on the Fe-doped surface of Fe-ReS₂. Subsequently, six consecutive protonation and reduction processes were canvassed as N_2 (g)* \rightarrow NNH* \rightarrow HNNH* \rightarrow $HNNH_2^* \rightarrow H_2NNH_2^* \rightarrow NH_2^* + NH_3 \rightarrow NH_3$ (Table S4, Supporting Information). For the reaction pathway on ReS₂, the NH₃ desorption step is the rate-determining step during NRR process due to the highest energy barrier demand of 2.15 eV; While the transformation of $HNNH_2^* \rightarrow H_2NNH_2^*$ becomes the rate-determining step for the reaction pathway on Fe-ReS₂ with a relatively lower energy barrier demand of 0.68 eV. The results indicate that the Fe-doped surface of Fe-ReS₂ is beneficial to providing more active sites for highly kinetic NRR process.

3. Conclusions

In summary, we demonstrate that modulating the energy level of electrocatalysts by low-valent transition-metal doping engineering is an effective strategy to improve the NRR and HER activity of ReS_2 based catalysts. By means of electrochemical analysis, in situ ATR IR, and DFT studies, we show the introduction of low-valent Fe motifs into ReS_2 structures results in the formation of abundant anchor sites for direct adsorption and subsequent activation of N₂ molecules, which are responsible for the significantly improved NRR and HER intrinsic activities. Our work unravels the important role of low-valent transition-metal on energy level refining for electrocatalysis optimization, which may also be used in many other catalytic systems.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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Conflict of Interest

The authors declare no conflict of interest.



Keywords

electrocatalysis, nitrogen reduction reaction, rhenium disulfide, transition-metals

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